## Design of High-Strength, High-Conductivity Copper Alloys through Controlling the Precipitation Behavior based on First-Principles Calculations

We investigated precipitation control in Cu-Ti alloys using first-principles calculations to design high-strength, high-conductivity copper alloys. DFT analysis of 22 elements revealed that P and Cr can enhance the cohesive energy of a-Cu<sub>4</sub>Ti, reduce its precipitation activation energy, and refine its precipitate size, thereby promoting a-Cu<sub>4</sub>Ti precipitation. These insights provide guidance for tailoring precipitation behavior in the development of advanced Cu-Ti alloys.

Cu-Ti alloys can reach tensile strengths above 1 GPa, which makes them promising alternatives to Cu-Be alloys [1]. Aging causes spinodal decomposition, forming metastable a-Cu<sub>4</sub>Ti (I4/m) as the strengthening phase, later transforming into  $\beta$ -Cu<sub>4</sub>Ti (Pmmn) [2,3]. Due to slow precipitation, optimizing strength and conductivity is difficult. Precipitation behavior depends on cohesive and interfacial energies of a-Cu<sub>4</sub>Ti with the Cu matrix. Previous DFT work on 3d transition metals [4] examined these separately but not their combined effect.

Here, we calculate these energies for a-Cu<sub>4</sub>Ti doped with 22 alloying elements (Li, Mg, Al, Si, P, Sc, V, Cr, Mn, Fe, Co, Ni, Zn, Ga, Ge, Y, Zr, Nb, Mo, In, Sn, Hf), derive precipitation activation energy and critical length, and identify elements that promote finer precipitation. First-principles DFT with PAW and PBE in VASP was used. Supercells for bulk a-Cu<sub>4</sub>Ti and two major interfaces with Cu were fully relaxed. Alloying elements were substituted into Cu or Ti sites at 6.25 at.%.

Twelve elements (P, Cr, Mn, Fe, Co, Ni, Mo, Li, Sc, Y, Zr, and Hf) increased cohesive energy ( $E_{coh}$ ), stabilizing a-Cu<sub>4</sub>Ti. Mn was effective at both sites. Interfacial energies ( $E_{int}$ ) were computed for Cu (001) // a-Cu<sub>4</sub>Ti (001) and Cu (310) // a-Cu<sub>4</sub>Ti (100). Among twelve dopants enhancing  $E_{coh}$ , only P and Cr lowered  $E_{int}$  at Cu (001) // a-Cu<sub>4</sub>Ti (001) interface by relieving lattice mismatch, but increased it at Cu (310) // a-Cu<sub>4</sub>Ti (100).

Total energy curves combining  $E_{coh}$  and  $E_{int}$  were analyzed against precipitate size. Figure 1, for Cr-doped a-Cu<sub>4</sub>Ti, shows reduced precipitation activation energy ( $E_a$ ) and critical length ( $L_c$ ) compared with undoped alloy. P and Cr lowered  $E_a$  by 21%

and 37% and reduced  $L_c$  by 11% and 26%, forming smaller, more uniform precipitates. Mn, Co, and Ni increased  $E_a$  but reduced  $L_c$ , indicating some refinement effect.

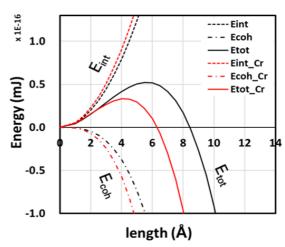


Fig. 1 The cohesive, interfacial and total energies for a-(Cu<sub>0.94</sub>Cr<sub>0.06</sub>)<sub>4</sub>Ti as a function of the length of precipitate. The black line represents the energies of the undoped one.

In conclusion, P and Cr are the most effective additives, stabilizing a-Cu<sub>4</sub>Ti and enhancing precipitation, offering a pathway to high-performance Cu-Ti alloys with tailored microstructures.

## <u>References</u>

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